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Research Article

Evaluation of the Particle Aerosolization from n-TiO₂ Photocatalytic Nanocoatings under Abrasion

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A parametric study on the release of titanium dioxide (TiO₂) nanoparticles from two commercial photocatalytic nanocoatings is carried out. For this, abrasion tests are performed on them. The formed aerosols are characterized by their number concentration, particle size distribution, individual particle shape, size, and chemical composition. The two nanocoatings appear to exhibit contrastingly opposite behavior with respect to the number concentration of the released particles. Having irregular shapes, the released particles are found to have unimodal size distributions with 1.5–3.5% (in mass) of Ti content. However, no free nanoparticles of TiO₂ were found. Distinct phases during the particle number concentration evolution with time are also discussed and evaluated. Two quantities— $(\Delta C/\Delta t)_I$ and T_{II} —are identified as the important indicators to qualitatively measure the resistance strength and hence the concentration of the released particles from a nanocoating during stress application.

1. Introduction

Photocatalytic nanocoatings are the special type of coatings that accelerate the reaction of forming activated oxygen from water or oxygen in the air. It accomplishes this formation by capturing ultraviolet light in the presence of photocatalyst titanium dioxide (TiO₂) nanoparticles [1, 2]. The formed activated oxygen is strong enough to oxidize and decompose organic materials and kill bacteria. As a result, these nanocoatings impart self-cleaning, air purifying, antibacterial, odor destructive, and super hydrophilic and antistatic (resistance of the static adsorption of small dust particles) properties to the surfaces on which they are applied. Such advantages have rendered their increasing use in construction or masonry applications like walls, pavements, and so forth [1–3]. However, during their lifecycle, these nanocoatings are subjected to various mechanical stresses in the form of the handling or processing of the parts coated with them [4–10]. This may result in their disintegration and the TiO₂ nanoparticles may start to get released in air in the form of aerosol [11, 12]. Upon exposure, these released TiO₂ nanoparticles may interact with the human organism

through inhalation or dermal contacts and get deposited inside the body. Various toxicological studies have demonstrated toxic effects of some types of TiO₂ nanoparticles in this case [13–15]. In spite of this, there is no sufficient information in the literature about the possible release of nanoparticles from photocatalytic nanocoatings. To better understand this, nanoparticle aerosol release evaluation is critical.

The study presented here aims to evaluate aerosol particle release or aerosolization from two commercial photocatalytic nanocoatings having nanoparticles of TiO₂. For simulating the stress conditions to which such coatings are subjected to, their abrasion is done. An experimental set-up is developed where the particles, after getting generated from abrasion, are characterized by aerosol measurement devices both in qualitative and semiquantitative ways.

2. Materials and Method

2.1. Surface Coating Material. For the study, two different commercially available photocatalytic nanocoatings were chosen. The details on the material properties are provided in

TABLE 1: Material properties of the two nanocoatings (data provided by the manufacturers).

Properties	Nanocoating 1	Nanocoating 2	Uncoated brick
Main composition	Nanotitanium Dioxide	Nanotitanium dioxide	Al, Si, Ca
Crystallite structure	Anatase	Anatase	—
Average primary particle size	<8 nm	<40 nm	7 μm (r.m.s. roughness)
Coagulation index	~2	~4	—
Appearance	Pale white liquid	Yellowish transparent liquid	Yellow
Dispersant	Water-based	Water-based	—
TiO ₂ vol. %	1.1	1	—

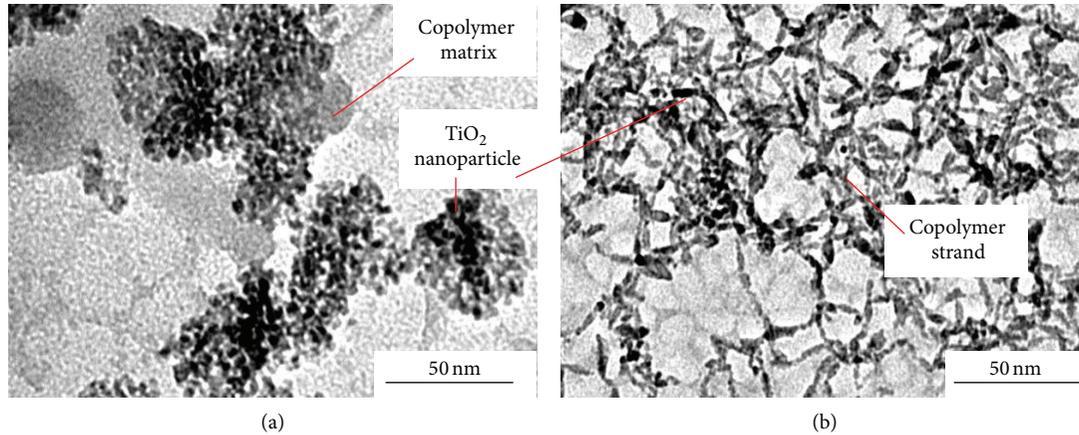


FIGURE 1: Microscopic analyses of the morphology of the nanoparticles present in the (a) nanocoating 1 and (b) nanocoating 2.

Table 1. The substrate chosen for the nanocoating application was a masonry brick (11 cm \times 5 cm \times 5 cm; Leopard brick, Ref: 901796, Castorama). It is basically an aluminosilicate brick which is frequently used in constructing façades, house walls, stairs, and so forth.

To evaluate the elemental composition and the constituent nanoparticles' morphology of the two nanocoatings, a transmission electron microscopy (TEM; Model CM12; Philips, The Netherlands) analysis was done. For this, two separate 1% (by volume) diluted solutions of the two nanocoatings were prepared. One drop (8 μL approximately) was taken from each diluted solution and was deposited on TEM copper mesh grids (Model S143-3; Quantifoil Micro Tools GmbH Germany). These mesh grids were made hydrophilic by their plasma treatment—0.1 mbar, 45 mA, 3 min—(Model K100X, Glow Discharge, Emitech, Quorum Technologies Ltd. UK) prior to drop deposition. After the deposition, the grids were then allowed to dry in a closed chamber so that the water content gets evaporated and the TiO₂ nanoparticles rest deposited on the grid.

In Figures 1(a) and 1(b), TEM images of the dried grids are shown. In Figure 1(a), the two phase agglomerates of the deposited particles from nanocoating 1 can be observed. These two phases are believed to be contributed by the copolymer (in grey color) and TiO₂ nanoparticles (in pitch black color). The average TiO₂ particle size is measured to be 8 ± 4 nm. However, in Figure 1(b), a network of stranded like structures from nanocoating 2 can be seen in which TiO₂ nanoparticles (appearing as small black chunks) are believed

to be embedded in copolymer strands. With a two phased structure, the average TiO₂ particle size is measured to be 25 ± 17 nm. The energy-dispersive X-ray analysis (EDX; Model X-max; Oxford Instruments UK) of the elemental composition of both nanocoatings shows similar chemical compositions, that is, C (60 to 65% in mass), O (15 to 20% in mass), Ti (10 to 15% in mass), Si (0 to 2% in mass), and traces of Al (<1% in mass).

The substrate surface preparation and the nanocoating application were done according to the technical data sheet recommended by the nanocoating manufacturers (i.e., degreased using brush and ethanol soaked paper, dry, and dust free surfaces; use of a high volume low pressure spray during coating; 25°C of ambient temperature). Different coating samples were prepared with two and four layers of both nanocoatings.

2.2. Abrasion Process. The standard Taber linear abrasion apparatus (Model 5750; Taber Inc. USA) was used for the abrasion of the nanocoatings. The apparatus is referenced in numerous internationally recognized test standards [16–18]. This apparatus is already being used in industries for analyzing the performance of products like paint, coating, metal, paper, textile, and so forth, during the application of a mechanical stress [19]. The stress being applied through Taber also corresponds to the typical one applied to surface coatings in a domestic setting, for example, walking with shoes and displacement of different objects [4, 7]. It incorporates

TABLE 2: Experimental parameters (repetition of the tests: 3 times).

Test sample	Number of layers of nanocoating (NOL)	Normal load			Abradant	Stroke length (mm)	Abrasion speed (cpm)	Number of cycles
		6 N	7.5 N	10.5 N				
Uncoated (reference)	0	X	X	X				
Nanocoating 1	2	X	X	X	H38	76	60	600
	4	X	X	X				
Nanocoating 2	2	X	X	X				
	4	X	X	X				

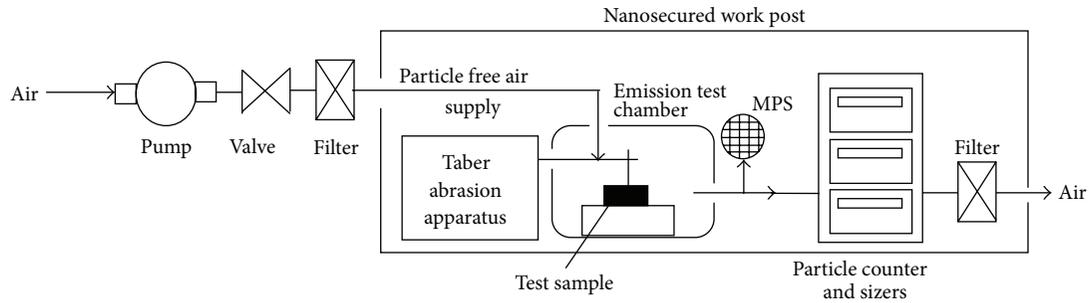


FIGURE 2: Experimental set-up.

a motor driven horizontal arm/bar that displaces an abradant in a back and forth linear motion over the test sample. The abrasion is caused by the friction at the contact surface between the surfaces of the abradant and the sample. Via a vertical shaft, a known weight is mounted on the top of the abradant which shall be referred to as the *Normal Load* in the text. The abrasiveness can be varied by changing the type of the abradant and normal load value. It is imperative to apply a reproducible and standardized stress on the test samples for reproducible tests. This has been ensured through preliminary tests and contextual literature survey on the optimal adjustment of the Taber abrasion apparatus [20–24].

2.3. Parameters Studied. In total, the effect of three experimental parameters on the concentration of released aerosol particles was studied: type of the nanocoating (i.e., nanocoatings 1 and 2), normal load (i.e., 6, 7.5, and 10.5 N), and number of layers of the nanocoating (i.e., 2 and 4).

The details can be seen in Table 2. An uncoated brick sample was also used for the reference. A Taber H38 nonresilient vitrified clay-carborundum abradant was used during the whole study [25]. The abrasion stroke length, abrasion speed, abrasion duration, and the number of abrasion cycles were kept constant at 76 mm, 60 cycles per minute (cpm), 10 min, and 600, respectively.

2.4. Experimental Set-Up. Figure 2 shows the scheme of the complete experimental set-up. Particle free air is passed through a nanosecured work post (HPPE 10, Erma Flux S.A., France) [26] containing the Taber abrasion apparatus. This work post has a particle filter efficiency of 99.99%. The test sample is placed inside a self-designed *Emission Test Chamber* (0.5 m × 0.3 m × 0.6 m) [27]. A slot is provided

on one of the walls of this chamber allowing the horizontal arm of abrasion apparatus passing through and operating the apparatus (equipped with motor) to be placed externally. The sampling of the generated aerosol particles is done in a close proximity of the test sample. The Taber apparatus along with the emission test chamber constitute aerosol generation section. The aerosol particles, getting generated during the abrasion process, are characterized in terms of their number concentration and number size distribution. The aerosol generation section is combined with an aerosol measurement section. It consists of a condensation particle counter (CPC; Model 3775; TSI Inc. USA; measurable size range: 4 nm to 3 μm), a scanning mobility particle sizer (SMPS; Model DMA 3081 and CPC 3775; TSI Inc. USA; measurable size range: 15 to 500 nm), an aerodynamic particle sizer (APS; Model 3321; TSI Inc. USA; measurable size range: 0.5 to 20 μm), and a miniparticle sampler (MPS; Ecomesure Inc. France) [28, 29]. The operation conditions of these instruments are as follows: CPC-flow rate of 1.5 L/min with 1 s of sampling time; SMPS-flow rate of 0.3 L/min with 120 s of sampling time; APS-flow rate of 5 L/min with 5 s of sampling time.

A MPS is used for the particle collection through filtration technique on copper mesh grids which can be used later in TEM for various qualitative analyses of the released particles without any limitation on the aerosol size.

Therefore, the whole aerosol measurement section, quantifying the particle release, can measure aerosol particles having sizes ranging from 4 nm to 20 μm. The deposition probability of this particle size range varies from 20 to 90% once they enter inside a human body [30]. The whole experimental set-up, in general, follows the ones used in [4, 31, 32]. For the analysis of the surfaces, a scanning electron microscope (QUANTA ESEM 400, FEI Inc., The Netherlands) has also been used.

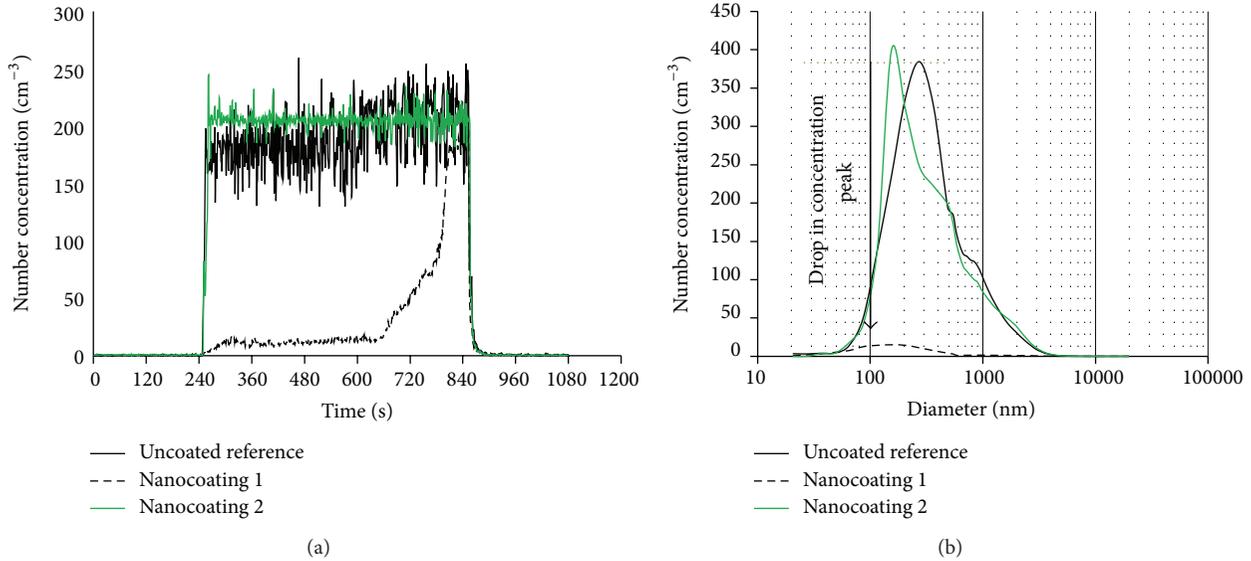


FIGURE 3: (a) Particle number concentration variation with time. (b) Size distribution of the aerosol particles generated during abrasion of 4 layered nanocoating samples under 6 N of normal load (note: all the curves are mean curves obtained from 3 repeated tests).

2.5. Background and Particle Loss. Three empty test runs were done before the real experiment to measure the concentration of the background particles and those generated by the abrasion apparatus. The abrasion apparatus was made to operate without the sample present in the chamber. The average concentration detected by CPC was $\sim 0.7 \text{ cm}^{-3}$ with a standard deviation of 0.2 cm^{-3} in the whole volume. Therefore, the concentrations of all the background particles and those generated by the abrasion apparatus were insignificant. The calculations of the loss of particle concentration due to their deposition on the walls of the chamber have shown a loss of 4% in number during 10 min (equal to the duration of the abrasion process).

Along with some turbulence losses, there can be some particle loss in the transport tubes also. These losses have not been taken into account while calculating the particle number concentrations. Hence, the present study is rather semiquantitative.

3. Results

3.1. Effect of the Type of Nanocoating. Figures 3(a) and 3(b) compare, through an example, the particle number concentration curves produced when the 4 layered nanocoating test samples were abraded under a normal load of 6 N. The test was repeated thrice under same conditions. For uncoated reference, the repetition was done on the same brick. In Figure 3(a), the abrasion starts at $t = 240 \text{ s}$ and ends at $t = 840 \text{ s}$. Before and after this time interval ($t = 0$ to 240 s), the system is idle.

The nanocoating 2 seems to impart no difference on the aerosol particle release when it is compared with the uncoated reference. The two have almost the same concentration levels. Since the nanocoating 2 probably gets rubbed off completely

without providing any resistance, the particle number concentration attains its maximum value ($C \approx 200 \text{ cm}^{-3}$) soon after the abrasion starts. The standard deviation ranges from 5 to 16 cm^{-3} .

For nanocoating 1, the number concentration is initially low ($C \approx 14 \text{ cm}^{-3}$) due to a probable resistance of the nanocoating 1 against abrasion. However, this resistance continues up to a certain point ($t = 624 \text{ s}$) after which the nanocoating 1 may start getting rubbed off. As a result, the number concentration starts increasing gradually. It attains the same value as for nanocoating 2 or the reference towards the end of the abrasion. The standard deviation in the values measured for nanocoating 1 varies from 0.7 to 27 cm^{-3} .

In Figure 3(b), the particle size distribution of the released aerosol particles, during the abrasion process, is shown. The nanocoating 2 seems to have no effect on the size distribution too. However, there is a shift of the size mode towards smaller particle sizes ($154 \pm 10 \text{ nm}$). The standard deviation in the particle size distribution values measured for nanocoating 2 changes from 0.2 to 16 cm^{-3} .

The nanocoating 1 considerably drops the peak of the number concentration curve by a factor of ~ 30 rendering the particle release totally insignificant. The standard deviation measured here is 8 cm^{-3} maximum.

In Figures 4(a) and 4(b), general overviews of the released particles are shown which were sampled during first 4 minutes of the abrasion test. A polydispersed aerosol from both nanocoating 1 (Figure 4(a)) and nanocoating 2 (Figure 4(b)) can be seen on the mesh grid with a high degree of agglomeration. A closer look on the morphology (Figure 4(c) for nanocoating 1 and Figure 4(d) for nanocoating 2) shows irregularly shaped aerosol particles having size ranging from hundreds of nanometer to tens of micrometer. The chemical analysis of these aerosol particles was found to have a Ti content of 1.5–3.5% (in mass) for both nanocoatings. However,

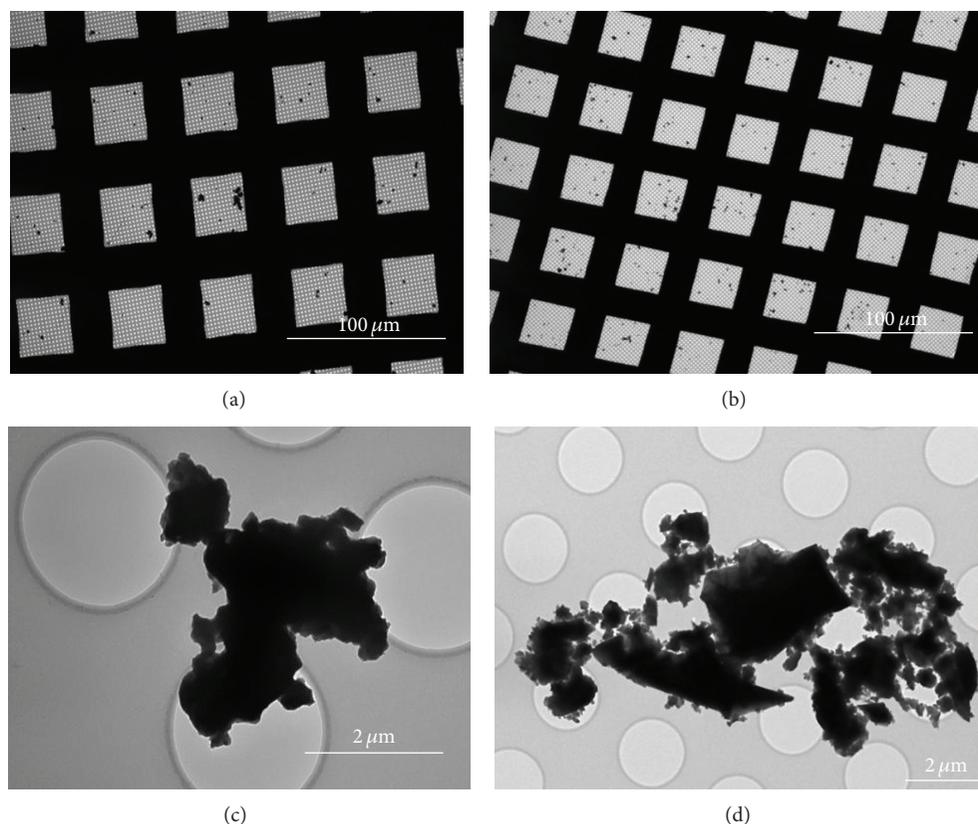


FIGURE 4: TEM image of the aerosol particles generated from (a) and (c) nanocoating 1 and (b) and (d) nanocoating 2.

the released aerosol particles from nanocoating 2 were found to be dominated by Si (50–70% in mass; essentially coming from the brick and the abrasant), C (5–7% in mass; essentially coming from the nanocoating 2 copolymer), and Al (10–20%; essentially coming from the brick). In case of nanocoating 1, Al was completely missing from the elemental composition of the released aerosol particles and the C content was elevated by 4–5%.

Therefore, from these observations, one can say that in case of nanocoating 1, the release of the aerosol particles was entirely contributed by nanocoating 1 itself. No particles came from the brick underneath. But for nanocoating 2, presence of the Al shows that the aerosol particles came from the brick too after the nanocoating 2 deterioration. It should be noted that for both nanocoatings, no free particles of TiO_2 were found on the grid. In fact, they were found to be embedded inside the copolymer matrix of the nanocoatings.

3.2. Effect of the Normal Load. In Figure 5(a), effect of increasing normal load has been shown on a 4 layered nanocoating 1 sample. The abrasion commences at $t = 240$ s and ends at $t = 840$ s. For a clear view of the particle number concentrations, between $t = 240$ s and $t = 480$ s, a zoomed view in Figure 5(a) is also shown. The released aerosol particle number concentration is found to be increasing with normal load. The same pattern continues in Figure 5(b) for a 4 layered nanocoating 2 sample too.

While measuring the particle size distribution in case of the nanocoating 1, the SMPS and APS showed very low concentrations which were even close to their particle detection thresholds. Hence, the two particle sizers were not employed further. But for nanocoating 2, there were no such problems. Thus, the size distribution, for nanocoating 2, is shown in Figure 5(c). Three unimodal distributions with increasing size modes (i.e., 154 nm to 274 nm to 365 nm) and increasing concentration peaks can be seen for increasing normal loads.

3.3. Effect of the Number of Layers. The number of layers also has a substantial effect on the aerosol particles generation. Figure 6 demonstrates this effect where two samples, having 2 and 4 layers of nanocoating 1, are tested under a normal load of 6 N. The abrasion commences at $t = 240$ s and ends at $t = 840$ s. The released particle number concentration is always lower when a 4 layered sample (std. deviation: 2 to 27 cm^{-3}) is abraded as compared to the two layered one (std. deviation: 13 to 37 cm^{-3}) or an uncoated reference. Both sets of layers seem to provide resistance towards abrasion. The SEM observations of the 4 layered nanocoating 1 sample were also done at the end of the abrasion. Figure 7 shows the observation. An unabraded coated surface (marked A) had an average Ti content of $\sim 12\%$ (in mass). For the abraded part (marked B), the average Ti content lowers down to $\sim 0\%$ (in mass), thus, completely exposing the brick surface. However,

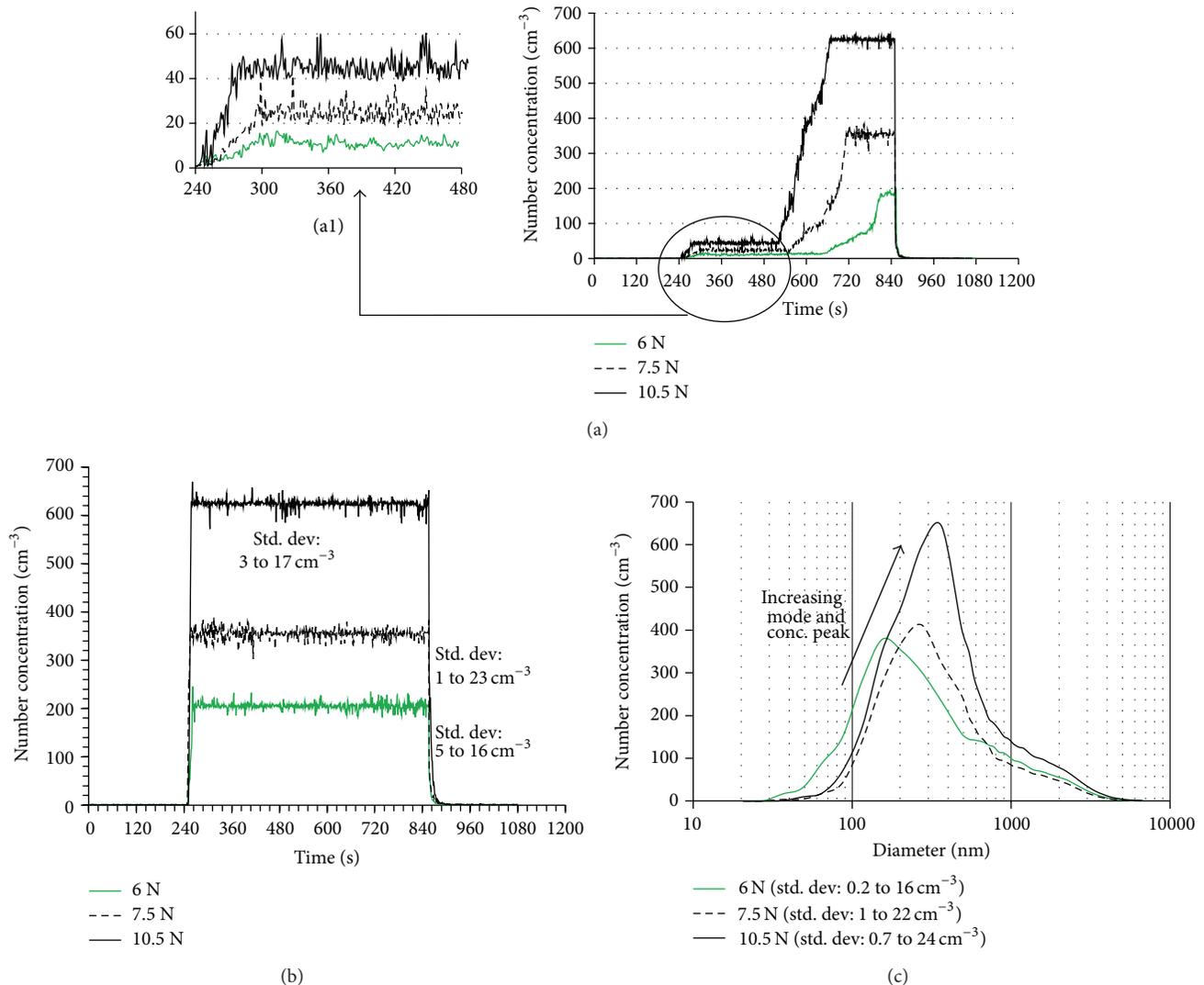


FIGURE 5: (a) Particle number concentration variation with time for 4 layered nanocoating 1 and (b) nanocoating 2 sample; (a1) zoomed view; (c) size distribution of the aerosol particles generated during abrasion of 4 layered nanocoating 2 sample (note: all the curves are mean curves obtained from 3 repeated tests).

in the case of nanocoating 2, both 2 and 4 layered samples had similar particle number concentrations.

4. Discussion

The particle number concentration ($\#C$) variations with time (t), shown in Figures 3(a) and 5(a), can be modeled in the forms shown in Figure 8 where Figure 8(a) corresponds to the uncoated reference, Figure 8(b) corresponds to the nanocoating 1, and Figure 8(c) corresponds to the nanocoating 2. Considering Figure 8(b), the curve for particle number concentration variation is constituted of 5 segments marked as EF, FG, GH, HI, and IJ.

The abrasion process starts at point E. As a result, $\#C$ starts rising from point E and stops at point F. This phase can be termed as *phase I*. During this phase, the contact surface conditions between the abrasant and the nanocoating are

believed to evolve due to changing surface roughness during abrasion. This evolution (marked as $(\Delta C_1/\Delta t)_I$) continues until the two surfaces come in equilibrium with each other which corresponds to point F. From the point F to the point G, $\#C$ remains constant, that is, *phase II*. During this phase, the abrasion of the nanocoating is being done under an equilibrium or stationary state and the nanocoating is still in a stable state. The EDX analysis of the nanocoating surface during this phase showed a strong presence of 5–8% (in mass) of Ti. There is some Ti content (~3.5% (in mass)) with no Al content in the sampled aerosol particles too (Figure 4(c)). From point G, $\#C$ rises again to the point H, that is, *phase III*. In this phase, the deterioration of the nanocoating is believed to start. The duration of this phase is decided by the level of the resistance a nanocoating can provide against its deterioration. Beyond point H, $\#C$ gets saturated and drops at point I (*phase IV*), where the abrasion stops. The nanocoating

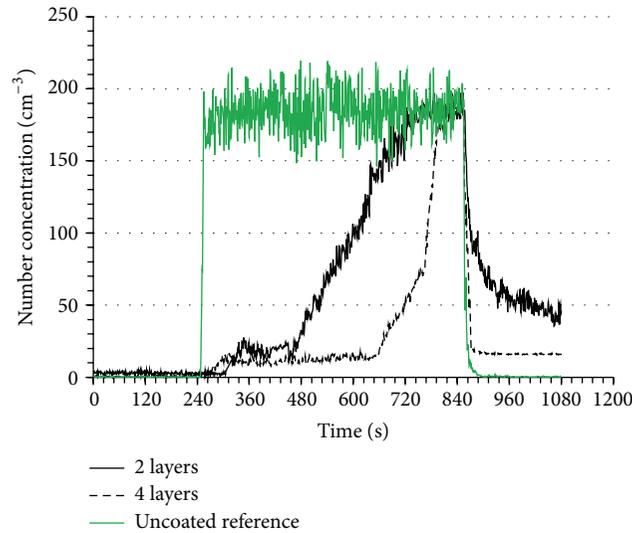


FIGURE 6: Particle number concentration variation with time for 2 and 4 layered nanocoating 1 samples (note: all the curves are mean curves obtained from 3 repeated tests).

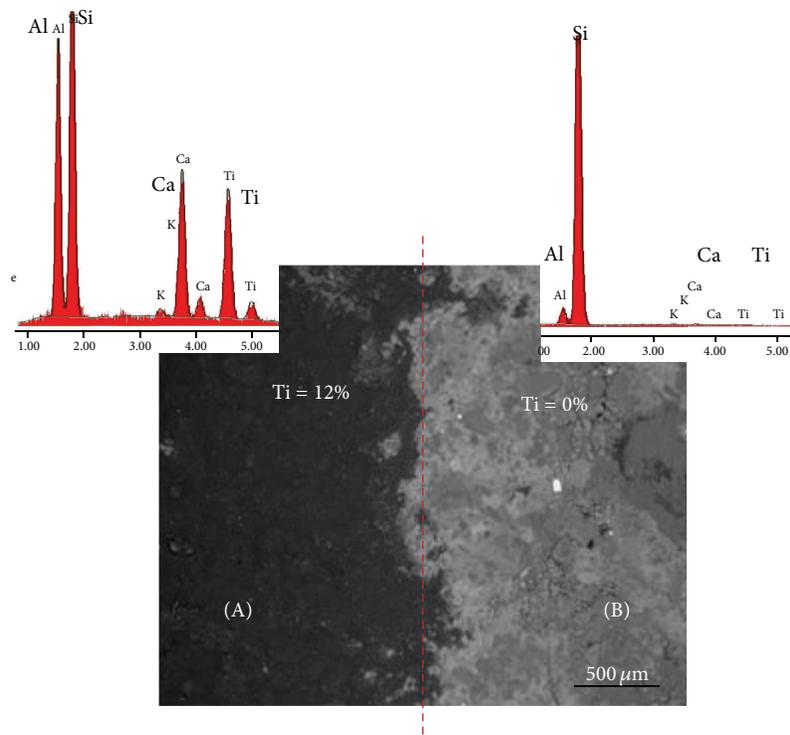


FIGURE 7: SEM image and EDX analysis of the coated and abraded parts of the nanocoating 1 sample; part (A): unabraded coated surface; part (B): abraded.

is supposed to be completely deteriorated during *phase IV* and the brick surface is exposed. The SEM of the brick surface done during this phase (Figure 7) shows a complete absence of Ti.

From these interpretations, since nanocoating 2 does not provide any resistance against abrasion and gets rubbed off easily, it appears that the *phases II* and *III* are totally absent

for nanocoating 2 and uncoated sample and longer *phases I* and *IV* seem to compensate for their absence. The same can be seen in Figures 8(a) and 8(c). If the subscript “0” signifies the uncoated reference, subscript “1” signifies the nanocoating 1 and subscript “2” signifies nanocoating 2, then one can deduce Table 3 where 8 segments (from Figure 8) are represented in the terms of 8 *concentration measures*.

TABLE 3: Representation of segments from Figure 8 in terms of concentration measures.

Segment	AB	BC	EF	FG	GH	HI	KL	LM
Representation	$\left(\frac{\Delta C_0}{\Delta t}\right)_I$	$(C_0)_{IV}$	$\left(\frac{\Delta C_1}{\Delta t}\right)_I$	$(C_1)_{II}$	$\left(\frac{\Delta C_1}{\Delta t}\right)_{III}$	$(C_1)_{IV}$	$\left(\frac{\Delta C_2}{\Delta t}\right)_I$	$(C_2)_{IV}$

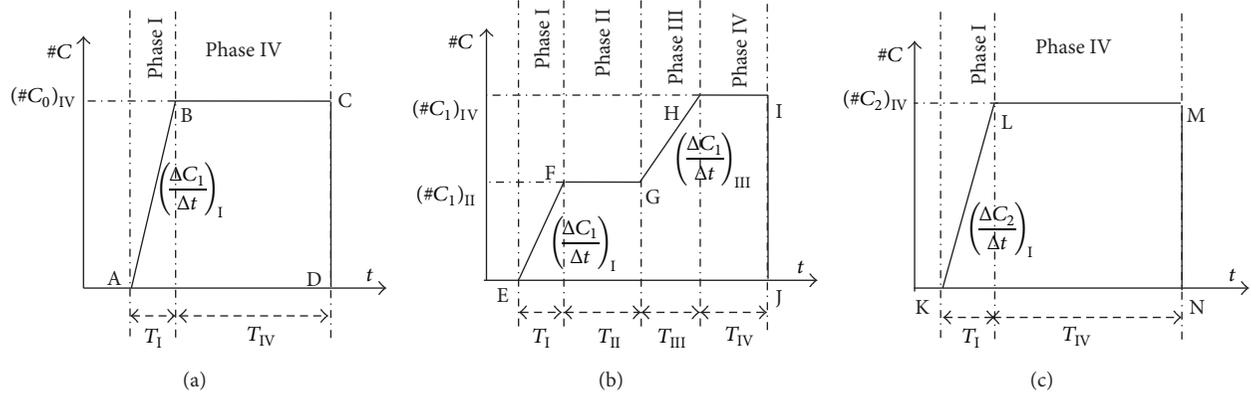


FIGURE 8: Generalized forms of the variation of aerosol particle number concentration generated from (a) uncoated reference (b) nanocoating 1 and (c) nanocoating 2.

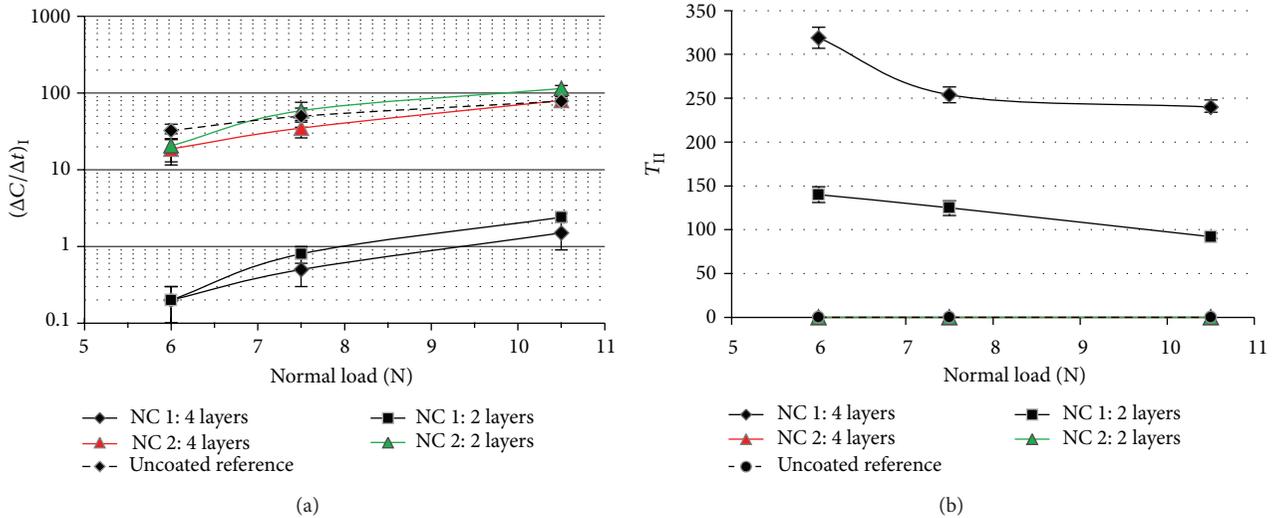


FIGURE 9: (a): Rate of change of the number concentration as a function of the normal load during phase I; (b) phase II duration as a function of the normal load for all nanocoating samples (note: all the curves are mean curves obtained from 3 repeated tests).

In Table 4, the values of these concentration measures are mentioned which were evaluated for all test samples and three normal load values. T_i is the duration of occurrence of the i th phase ($i = I, II, III, IV$). On the basis of this table, it can be said that irrespective of the normal load acting during abrasion, $(\Delta C_0/\Delta t)_I > (\Delta C_1/\Delta t)_I$; $(\Delta C_2/\Delta t)_I > (\Delta C_1/\Delta t)_I$; $(\Delta C_1/\Delta t)_I < (\Delta C_1/\Delta t)_{III}$; $(\#C_0)_{IV} \approx (\#C_1)_{IV} \approx (\#C_2)_{IV}$.

Based on the values obtained in Table 4, rate of change of the number concentration of the released particles during phase I, $(\Delta C/\Delta t)_I$, can be plotted as a function of the normal load for all test samples. This is shown in Figure 9(a). Similarly, the duration of the phase II (T_{II}) can also be plotted as a function of the normal load (see Figures 9(a) and 9(b)).

From these figures, the curves corresponding to the nanocoating 1 (NC 1: 2 layers and NC 1: 4 layers) are clearly distinct from those of the nanocoating 2 (NC 2: 2 layers and NC 2: 4 layers) and the uncoated reference. The nanocoating 1 samples hold the lowest values of $(\Delta C/\Delta t)_I$ with a factor of difference almost equal to 100. This demonstrates their high resistance towards abrasion. At the same time, the nanocoating 1 samples hold the highest values for T_{II} also which demonstrates the highest duration of the stability of the nanocoating 1 during abrasion. This stability increases with the increase in number of nanocoating layers and decreases with the increase in normal load. The nanocoating 2 and the uncoated reference samples have $T_{II} = 0$.

Therefore, the two quantities— $(\Delta C/\Delta t)_I$ and T_{II} —can be used as the *indicators* for the measurement of the particle release tendency of a nanocoating subjected to the abrasion. A more stable (i.e., high T_{II}) and lesser deterioration prone (i.e., low $(\Delta C/\Delta t)_I$) nanocoating yields lesser particle release. The T_{II} can be as high as the duration of the stress application (319 s in the present case) and $(\Delta C/\Delta t)_I$ can be as low as 0 ($0.2 \text{ cm}^{-3} \text{ s}^{-1}$ in the present case).

5. Conclusion

This study has investigated the possibility of the release of aerosol nanoparticles from two commercially available TiO_2 photocatalytic nanocoatings under mechanical stress conditions, simulated using an abrasion process.

The 4 layered nanocoating 1 sample has performed best in inhibiting the particle release, followed by the 2 layered one. However, the nanocoating 2 has not succeeded at all in its inhibition (Figures 3, 5, and 6).

The chemical analysis of the released aerosol particles has shown that, owing to a fast deterioration of the nanocoating 2, the particles were essentially contributed by the brick rather than the nanocoating 2 during abrasion. But in the case of nanocoating 1, it is the other way around.

No free nanoparticles of TiO_2 were found to be present among the released aerosol.

Four different phases during the particle release have been identified and evaluated for all three test samples (Figure 8, Tables 3 and 4).

The two particle release indicators— $(\Delta C/\Delta t)_I$ and T_{IV} —have been introduced which can be used for measuring the holding strength or particle release tendency of a nanocoating (Figure 9).

Further tests shall be done with other nanocoating samples and normal load values to develop a standard test procedure for measuring $(\Delta C/\Delta t)_I$ and T_{IV} . These tests shall be accompanied by other analytical tests too to further strengthen the support for the complete physical interpretations of the four phases.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

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